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KSbO(Ge_{0.32}Si_{0.68})O₄, a KTP isomorph

Stefan T. Norberg,* Göran Svensson and Jörgen Albertsson

Department of Inorganic Chemistry, Chalmers University of Technology, SE-412 96 Göteborg, Sweden Correspondence e-mail: stn@inoc.chalmers.se

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A structural model of potassium antimony germanate/silicate (0.32/0.68), KSbO(Ge_{0.32}Si_{0.68})O₄, has been determined at room temperature. KSbO(Ge_{0.32}Si_{0.68})O₄ belongs to the KTiOPO₄ (KTP) isomorphic family and is composed of SbO₆ octahedra (site symmetry $\overline{1}$ and 2) arranged in helical chains bridged by (Ge/Si)O4 tetrahedra. Germanium and silicon have a similar distribution in the crystallographically independent tetrahedra (site symmetry 2). The structure contains large cavities occupied by the K atom. Two partially occupied potassium positions have been identified 1.273 (8) Å apart, with an indication of a third potassium position between them. At room temperature, KSbO(Ge_{0.32}Si_{0.68})O₄ crystallizes in the paraelectric phase of space group Pnan. This phase is found at elevated temperatures for almost all KTiOPO₄ isomorphic compounds and $KSbO(Ge_{0.32}Si_{0.68})O_4$ is the second isomorph that is paraelectric at room temperature.

Comment

Since the structure determination by Tordjman et al. (1974) of potassium titanyl phosphate, KTiOPO₄ (KTP), it has become a well known material for use in non-linear optical processes (Zumsteg et al., 1976), such as second harmonic generation (SHG; Bierlein et al., 1989). The high non-linearity of KTP extends to many isomorphic compounds with the general chemical formula $AMOBO_4$ (A = K, Rb, Cs or Tl; M = Ti; B = P or As). Several of these isomorphs have been extensively studied (Stucky et al., 1989, and references therein). Recently, another group of isomorphs have been explored, with substitutions in the MO_6 octahedra and the BO_4 tetrahedra (A = Na, K, Rb, Tl or Ag; M = Sb, Nb or Ta; B = Si orGe). The isomorphs with Sb in the octahedra have been fairly well studied. KSbOSiO₄ (KSS) was prepared by Crosnier et al. (1990) in a solid-state reaction and the structure of KSbOGeO₄ (KSG) was published independently by Pagnoux et al. (1991) and Belokoneva et al. (1991). Other compounds include NaSbOSiO₄ (Pagnoux et al., 1992), RbSbOGeO₄ (Favard et al., 1992), TlSbOGeO₄ (Belokoneva & Mill, 1992a), RbSbOGeO₄ (Belokoneva & Mill, 1992b), NaSbOGeO₄, AgSbOGeO₄ and AgSbOSiO₄ (Mill *et al.*, 1993).

The KTP isomorphic materials with Sb in the oxygen octahedra all have weak non-linear optical coefficients, which are orders of magnitudes lower than those of KTP. This is mainly due to the low polarizability of Sb^V and the regularity of the SbO₆ octahedra. Butashin *et al.* (1994) reported that an exchange of Sb^V with Nb^V greatly increases the non-linear optical coefficients. The solid solution system of KSb_{1-x}Nb_xOGeO₄ extends to x = 0.75, but no KTP phase has been detected in the A_2 O–Nb₂O₅ BO_2 (A = Na, K, Rb, Tl or Ag; B = Si or Ge) system (Belokoneva *et al.*, 1991).

As a part of our ongoing investigation of new modifications and isomorphic materials in the KTP family, we have prepared a new member of the Sb family with mixed occupation of Ge and Si in the tetrahedral positions, viz. KSbO(Ge_{0.32}Si_{0.68})O₄ (KSGS).

The structure of KSGS is very similar to the KTP structure (Thomas *et al.*, 1990). It comprises SbO₆ octahedra that are corner linked at alternating *cis* and *trans* positions, forming helical chains in the [011] direction. These chains are further linked by (Ge/Si)O₄ tetrahedra. The framework built by the octahedra and tetrahedra has large cavities which are occupied by the K⁺ cations. Fig. 1 gives a view of the structure in the (010) plane.

All SbO₆ octahedra are fairly regular with maximum and minimum Sb–O bond lengths of 2.004 (5) and 1.955 (6) Å, respectively, and a mean (s.u.) of 1.97 (2) Å. The AO_4 tetrahedra (A = Ge, Si) are also regular with maximum and minimum bond lengths of 1.683 (6) and 1.648 (6) Å and a mean (s.u.) of 1.664 (15) Å. The mean (s.u.) Si–O bond length of the SiO₄ tetrahedra in KSS is 1.62 (2) Å (Crosnier *et al.*, 1990), while that of the GeO₄ tetrahedra in KSG is 1.742 (14) Å (Belokoneva *et al.*, 1991). It should also be noted that the shape and bond lengths of the SbO₆ octahedra are almost identical in KSS, KSG and the new KSGS. There is,





*ORTEP*III (Burnett & Johnson, 1996) view of the KSbO(Ge_{0.32}Si_{0.68})O₄ structure in the (010) plane. Displacement ellipsoids are drawn at the 80% probability level. [Symmetry codes: (i) 1 - x, $y + \frac{1}{2}$, $z + \frac{1}{2}$; (ii) 1 - x, $y - \frac{1}{2}$, $z + \frac{1}{2}$; (iii) $-x + \frac{3}{2}$, $-y + \frac{3}{2}$, $z + \frac{1}{2}$; (iv) 1 - x, 2 - y, 1 - z; (v) 1 - x, 1 - y, 1 - z; (vi) $-x + \frac{3}{2}$, y, 1 - z; (vii) $x, -y + \frac{3}{2}$, $-z + \frac{1}{2}$; (viii) 1 - x, $y + \frac{1}{2}$, $z - \frac{1}{2}$.]

however, one large difference between KSGS and the KSS and KSG structures, in that KSGS has split cation positions. The K and K' positions are separated by 1.273 (8) Å, each with an occupancy of about 0.50. Split cation positions have been reported earlier for the *A*SbOGeO₄ materials in the hightemperature paraelectric phase but with shorter distances between the split positions, *e.g.* 0.561 (5) Å for TlSbOGeO₄ at 293 K (Belokoneva & Mill, 1992*a*) and 0.703 (13) Å for RbSbOGeO₄ at 503 K (Belokoneva *et al.*, 1997).

The coordination sphere of O atoms around K and K' are irregular. K is coordinated by six O atoms, resulting in four short bonds [2.707 (8) < K-O < 2.866 (8) Å] and two longer interactions up to 3.112 (7) Å. K' has seven coordinating O atoms, four shorter bonds [2.659 (8) < K'-O < 2.734 (8) Å] and three longer interactions up to 3.146 (7) Å. Table 1 gives all bond lengths for K and K'. The four short K-O bonds at each split position (< 2.90 Å) are approximately in the (001) plane. A similar trend of irregular coordination around the potassium sites exists in KTP (Thomas *et al.*, 1990), but with eight coordinating O atoms for K1, interactions up to 3.057 (3) Å and nine coordinating O atoms for K2 within 3.117 (3) Å. Table 2 gives the unit-cell parameters for KSS, KSG and KSGS.

The high-temperature modifications (space group Pnan) of KSS and KSG were studied by Favard et al. (1994) at temperatures of $T_c = +30$ K. The distance between the split cation positions was 0.909 (13) Å in KSS and 0.843 (13) Å in KSG. It was noted that the deviation from centrosymmetry in the low-temperature phase (space group $Pna2_1$) was mainly due to the location of A^+ ions and not from the framework. The changes in the framework during the phase transition are small compared with KTP, mainly due to the regularity of the SbO_6 compared with the TiO₆ octahedra, in which the Ti^{IV} atoms are off-centre. The phase transition is displacive for the framework and of order-disorder type for the cations (Favard et al., 1994). The centrosymmetry of ASbO(Ge/Si)O₄ compounds arises from the partial but equal occupation of two non-equivalent cation sites, resulting in the paraelectric phase. At low temperature, the four partly occupied A^+ sites reduce to two fully occupied sites, resulting in an ordered structure with $Pna2_1$ symmetry.

All of those previously studied Sb-containing materials are isomorphic with KTP, with a ferroelectric to paraelectric phase transition well above room temperature (from space group *Pna2*₁ to *Pnan*). An exception is TISbOGeO₄ with a T_c of 272 K (Stefanovich *et al.*, 1993). The T_c of both KSS and KSG was determined as 600 K by Stefanovich *et al.* (1993). KSGS is the second material in the KTP family to have a T_c below room temperature. Preliminary investigations of KSGS at low temperatures (173 and 113 K) show that the space group is still *Pnan, i.e.* the value of T_c must be very low.

Experimental

The crystals were obtained in a platinum crucible by spontaneous crystallization in a PbO_2 flux containing equimolar amounts of K_2CO_3 , Sb_2O_5 , GeO_2 and SiO_2 carefully mixed together. The weight

Selected bond lengths (Å).

K-K'	1.273 (8)	K'-O1	2.659 (8)
K-01	2.707 (8)	$K' - O2^i$	2.679 (7)
K-O5 ⁱ	2.728 (7)	K'-O3 ⁱⁱ	2.700 (7)
K-O3 ⁱⁱ	2.854 (7)	$K' - O5^i$	2.734 (8)
K-O4	2.866 (8)	$K' - O4^{iv}$	3.036 (8)
$K - O2^i$	3.024 (7)	$K' - O1^{iv}$	3.133 (8)
K-O3 ⁱⁱⁱ	3.112 (7)	$K' - O5^{ii}$	3.146 (7)

Symmetry codes: (i) $\frac{3}{2} - x$, 1 + y, 1 - z; (ii) $\frac{1}{2} + x$, 1 - y, z; (iii) $\frac{1}{2} + x$, $\frac{1}{2} + y$, $\frac{1}{2} - z$; (iv) $\frac{3}{2} - x$, y, 1 - z.

Table 2

Unit-cell parameters for KSS, KSG and KSGS.

	a (Å)	b (Å)	c (Å)	$V(\text{\AA}^3)$
KSS	13.005 (1)	6.4748 (6)	10.614 (1)	893.7 (3)
KSG	13.224 (2)	6.597 (1)	10.759 (2)	938.6 (3)
KSGS	13.0970 (9)	6.5310 (5)	10.6873 (6)	914.15 (11)

ratio between PbO₂ and the crystallization material was 1:1. The mixture was heated slowly for 4 d to 1273 K in order to obtain a homogeneous melt and the temperature then decreased to 1023 K at a rate of 1.4 K h⁻¹. The brown flux was later dissolved in 5 *M* HCl. Most of the flux dissolved within 24 h. The small fine crystals in the resulting yellow powder had the typical KTP morphology (Bolt & Bennema, 1990). It may also be noted that mixing K₂CO₃, Sb₂O₅, GeO₂ and SiO₂ without any PbO₂ as a flux resulted in a powder (no melting occurred at 1273 K) containing small fine crystals of K₃Sb₅O₁₄, whose structure has been published by Hong (1974).

Crysta	l data
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2818 reflections

 $w = 1/[\sigma^2(F^2) + 0.01(F^2)^2]$

89 parameters

KSbO(Ge _{0.32} Si _{0.68})O ₄ $M_r = 283.06$ Drthorhombic, <i>Pnan</i> n = 13.0970 (9) Å b = 6.5310 (5) Å c = 10.6873 (6) Å V = 914.15 (11) Å ³ Z = 8 = 32000000000000000000000000000000000000	Mo K α radiation Cell parameters from 21 reflections $\theta = 35.1-39.6^{\circ}$ $\mu = 9.13 \text{ mm}^{-1}$ T = 293 K Prism, colourless $0.06 \times 0.05 \times 0.04 \text{ mm}$
$D_x = 4.118 \text{ Mg m}^{-5}$	
Enraf-Nonius CAD-4 diffract- ometer ω -2 θ scans Absorption correction: analytical (Alcock, 1974) $T_{\min} = 0.784, T_{\max} = 0.804$ 5155 measured reflections 2818 independent reflections 1585 reflections with $F^2 > 2\sigma(F^2)$	$R_{int} = 0.074$ $\theta_{max} = 39.8^{\circ}$ $h = 0 \rightarrow 23$ $k = 0 \rightarrow 111$ $l = -19 \rightarrow 19$ 3 standard reflections frequency: 240 min intensity decay: none
Refinement	
Refinement on F^2 $R[F^2 > 2\sigma(F^2)] = 0.046$ $\nu R(F^2) = 0.168$ S = 1.00	$(\Delta/\sigma)_{max} < 0.001$ $\Delta\rho_{max} = 3.41 \text{ e} \text{ Å}^{-3}$ $\Delta\rho_{min} = -3.92 \text{ e} \text{ Å}^{-3}$ Extinction correction: Gaussian

Wilson (1949) statistics clearly indicate a centrosymmetric structure. However, an attempt to refine the structure in space group $Pna2_1$ using the coordinates of KTP (Thomas *et al.*, 1990) as the starting set led to severe correlation problems. No indication of any twinning was found in the measured KSGS crystal. Independent

(Zachariasen, 1967)

Extinction coefficient: 1.9 (9) \times 10³

measurement of three different KSGS crystals from the same batch have each resulted in the same space group (*Pnan*). Careful examination of the reflection profiles of our CAD-4 diffractometer data shows nothing out of the ordinary and no extra peaks appear using an area detector.

The highest residual electron-density peak (based on the observed reflections) is located between the K and K' positions, while the lowest peak is 1.571 (6) Å from the K position. $\Delta\rho$ maps of the area around and between K and K' show a single larger peak positioned 0.655 (6) Å from K and 0.643 (6) Å from K'. The peak could be refined as a possible K' site to an occupancy factor of 0.07 using restraints. Because of the restraints we have chosen to neglect K' and report the structure with two split K positions as our final structural model.

An isotropic extinction parameter (Zachariasen, 1967) was refined using Larson's implementation (Larson, 1970). About 14% of the reflections were affected, with a maximum correction of y = 0.90 for the 022 reflection (the observed structure factor is $F_{\rm obs} = yF_{\rm kin}$, where $F_{\rm kin}$ is the kinematic value of the structure factor).

Data collection: *CAD-4 Software* (Enraf–Nonius, 1989); cell refinement: *LATCON* in *Xtal3.7* (Hall *et al.*, 2000); data reduction: *DIFDAT, SORTRF, ADDREF* and *ABSORB* in *Xtal3.7*; program(s) used to solve structure: *SHELXS97* (Sheldrick, 1997); program(s) used to refine structure: *CRILSQ* in *Xtal3.7*; molecular graphics: *ORTEP*III (Burnett & Johnson, 1996); software used to prepare material for publication: *BONDLA, ATABLE* and *CIFIO* in *Xtal3.7*.

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Supplementary data for this paper are available from the IUCr electronic archives (Reference: BR1312). Services for accessing these data are described at the back of the journal.

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